

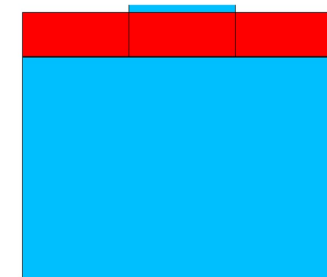
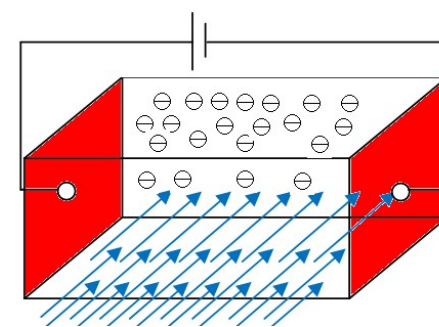
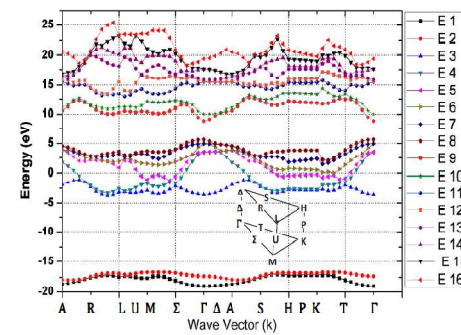
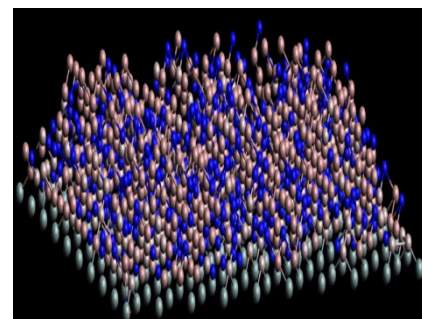


Horizontal MOCVD Injector Reactor

TNL- Injector Simulator



*Technology of Next Level
driven through innovation*



CHALLENGES: MOCVD EPITAXY



- ❑ MOCVD → *Chemical process* to deposit special materials and high-tech devices
- ❑ Growth mechanism Investigation of III–V compounds → *High value products*
- ❑ Film growth properties, closely related to product quality → Depends *gas inlet flow rate, operating pressure & temperature* and *reactor geometry configurations*
- ❑ High-quality demands of the semiconductor industry → control composition distributions in a MOCVD reactor to achieve *film growth thickness & uniformity*
- ❑ MOCVD : *complex transport phenomena* → momentum, heat, mass & chemical reactions
- ❑ MOCVD reactor's optimal growth condition → requires extra experimentation to refine & search process to locate an optimal solution until film deposition quality satisfies a pre-specified criteria.
- ❑ Two categories: → Development of a chemical kinetic & Transport phenomenon
- ❑ To save design cost & development time → use of *CAD methodologies* of MOCVD processes.

CHALLENGES: GaN/Si EPITAXY



- ❑ *GaN directly on Si* encounters several challenges → large lattice mismatch (16% to 20.4%) and thermal expansion coefficient (*TEC*) mismatch ~ 53% leads to the generation of defects at room temperature
- ❑ *AlN buffer layer* growth prior to the growth of GaN → To avoid formation of cracks and several other technological challenges
- ❑ *Stack of $Al_xGa_{1-x}N$ layers* on AlN buffer layer → To balance compressive and tensile strain
- ❑ Growth of the AlN buffer layer on the Si substrate → *Still not well understood*
- ❑ Well known phenomenon of formation of *amorphous SiNx*, deteriorates crystalline quality.
- ❑ Pre-nitridation process of Si substrate greatly influences surface morphology. However, nitridation time and substrate nitridation temperature highly influence strain generation between each layer and results in defect formation process
- ❑ NH_3 pre-flow with smaller step size time and the optimum substrate temperature are still *debatable* to achieve a single-crystalline GaN on Si substrate

CHALLENGES: GaN/Si EPITAXY



- ❑ GaN-on-Si devices recently attracted much attention *for radio frequency (RF) applications* due to *cost & large size Si substrates* with possibility of *co-processing in CMOS foundries*
- ❑ However, GaN-on-Si RF devices, still facing some challenges in form of **RF loss**
- ❑ Contributor to RF loss is *parasitic channel formation* at AlN/Si interface
- ❑ Type of conductivity & formation mechanism of parasitic channel is **controversial**
- ❑ Few groups suggest formation of a p-type conductive layer at the AlN/Si interface, while others suggest that an n-type electron channel induced by the strong polarization field at the AlN/Si interface.
- ❑ Understanding the behaviors of the parasitic layer is crucial for reducing the RF loss
- ❑ Unsolved problem requires an unambiguous identification of this conductive layer and correlating it with the RF loss

AVAILABLE SOLUTION



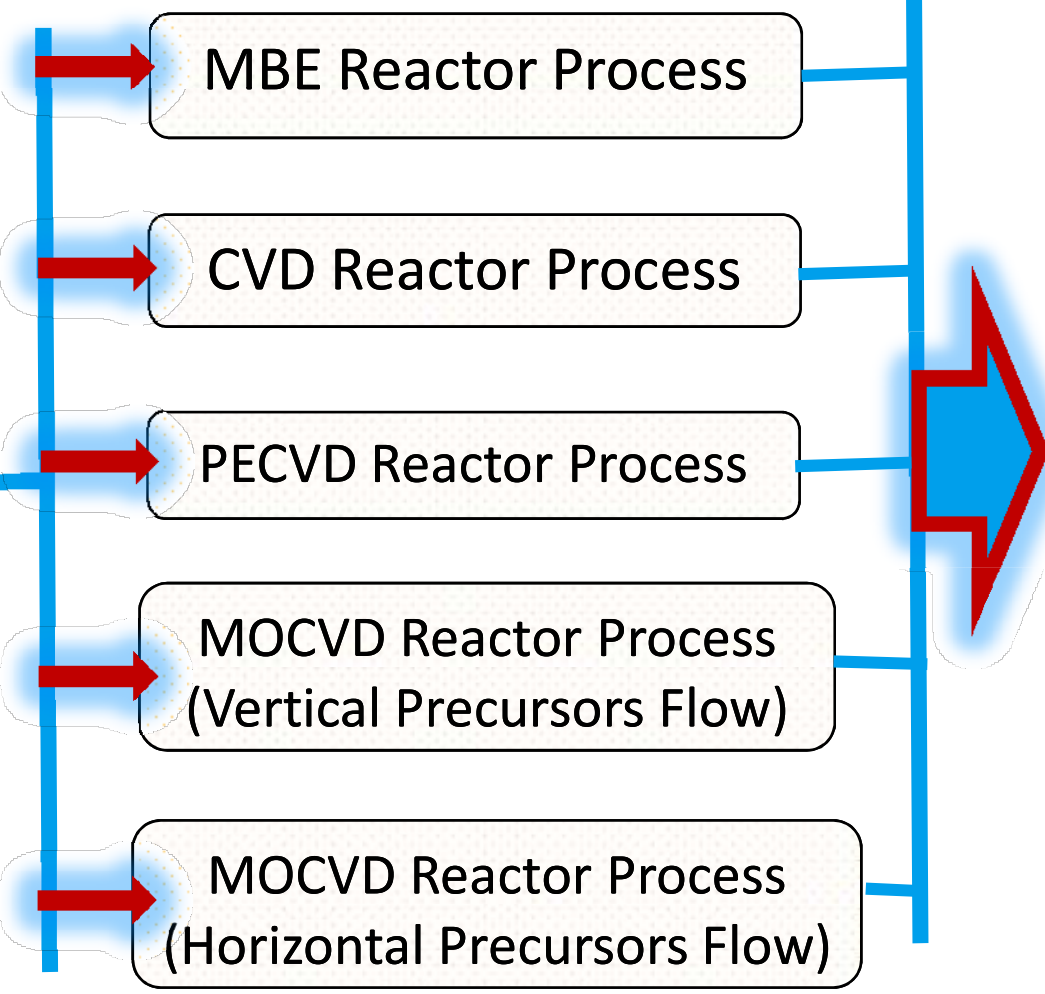
- ❑ Numerical simulations, (*CFD software*) *ONLY* predicts → Film growth rate and uniformity, Prediction accuracy heavily relies on the reaction kinetics mechanism, however, transport phenomena oversimplified *by assuming constant mass diffusivity for precursors, temperature-independent properties of gas mixture, and constant wall temperature profile or constant heat transfer coefficient for boundary condition of inner reactor wall.*
- ❑ ***Oversimplifications*** made for the heat transfer → substantially prohibit the prediction ability of the MOCVD processes, making a great discrepancy between the model and the actual behavior of the film growth on substrate.
- ❑ To extract atomistic scale information of MOCVD reactor process through CFD software is **IMPOSSIBLE**

EPITAXY PROCESS SOLUTION



Innovative Atomistic Scale Reactor Simulation without use of Continuum models

TNL Framework



- User's Input Growth Conditions
- Surface Profile (Roughness)
- Strain Mapping (layer by Layer)
- Lattice parameter
- Defects (Vacancies, Interstitials, Dislocations, Stacking Faults)



INPUTS : MOCVD PROCESS



Chamber Condition

Showerhead Based

Injector Based

Injector Parameters

| | | |
|-------------------------|----------------------------------|-----------------------------------|
| Chamber Volume (ltrs.) | <input type="text" value="1.4"/> | |
| Chamber Pressure | <input type="text" value="10"/> | <input type="text" value="torr"/> |
| Ceiling Height (cm) | <input type="text" value="2.0"/> | |
| Chamber Temperature (C) | <input type="text" value="100"/> | |
| Sticking Coeff. | <input type="text" value="1"/> | |

Many More parameters details Require

Precursor Condition

Number of Port

Precursor 1

Flow Rate

Step 1

Precursor Condition

Number of Port

Precursor 1

Flow Rate

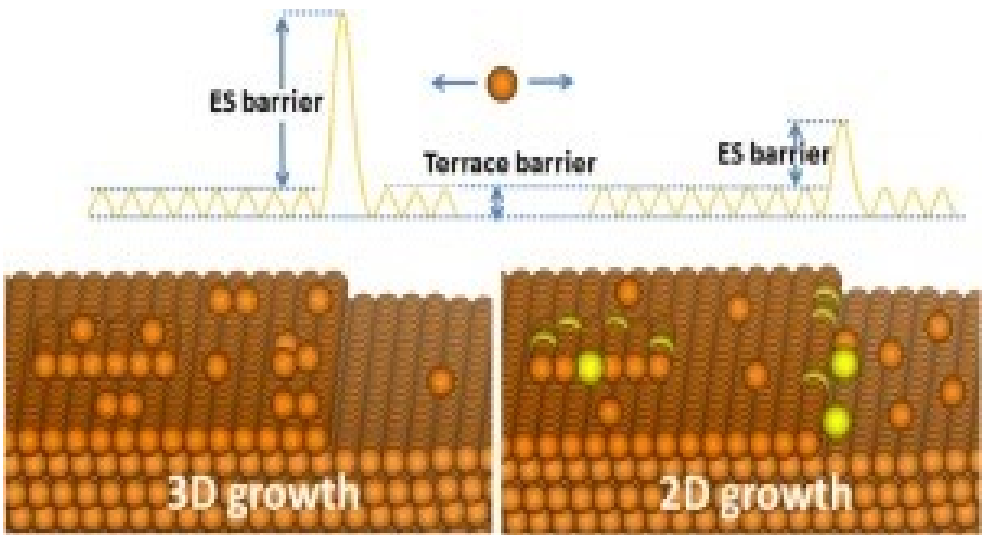
- Select Precursor
- Ga(CH3)3
- (CH3)3CAsH2
- H2
- Al(CH3)3
- NH3
- CH3
- O2



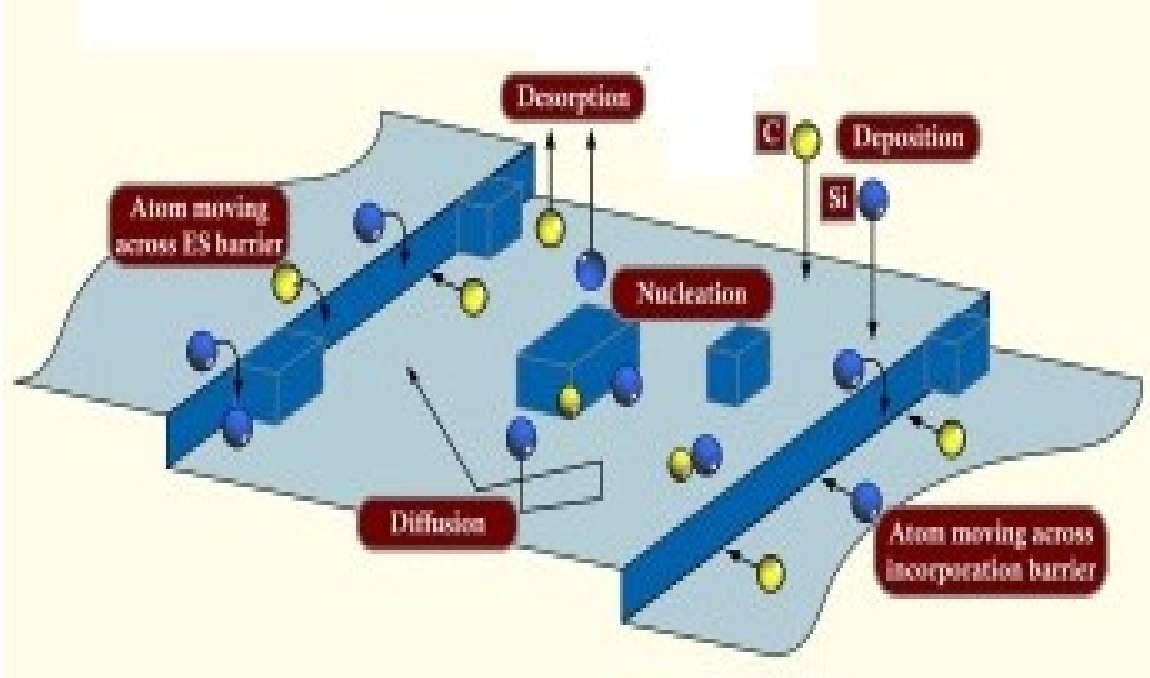
METHODOLOGY: MOCVD PROCESS



Schwoebel barrier: The atom diffuses from the site exactly above the edge atom to the site immediately next to the edge atom as;



Incorporation barrier: The atom incorporates into the edge on the same surface level.

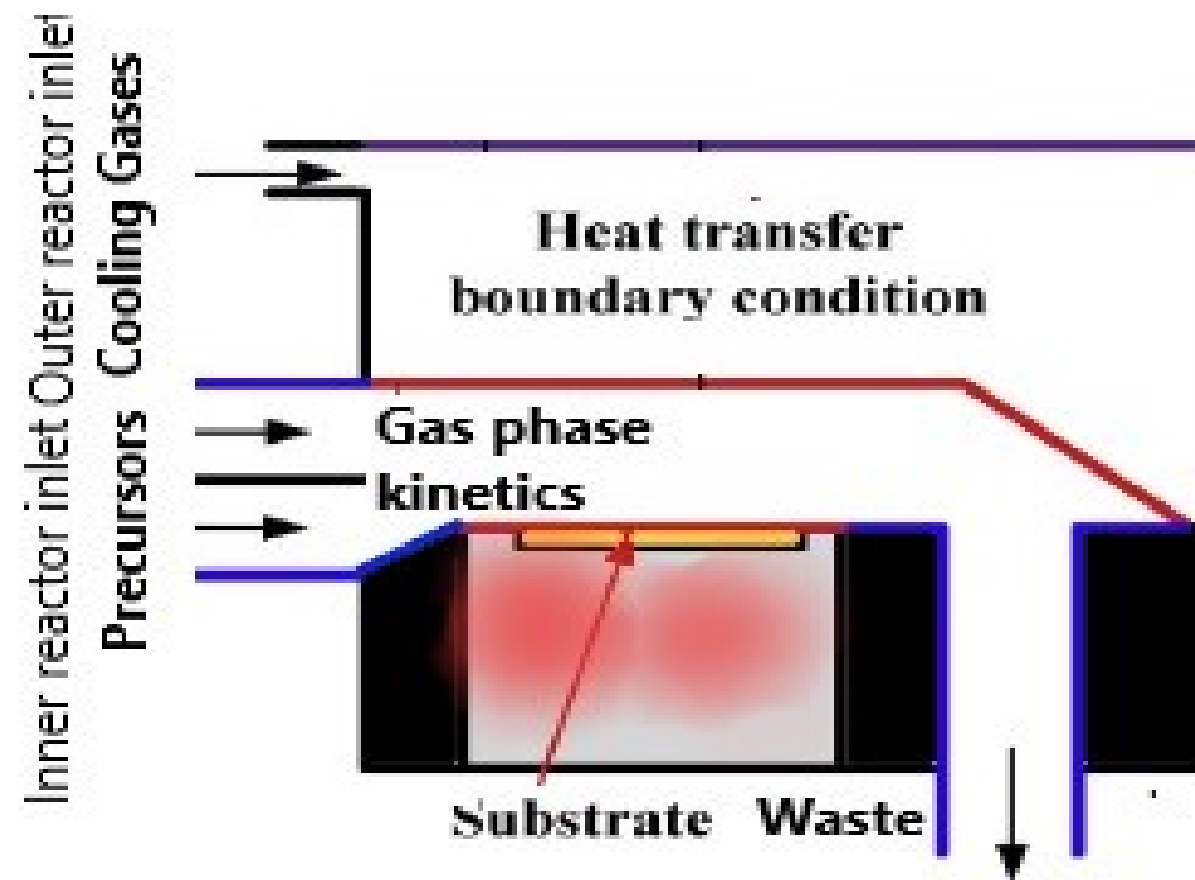


PLANETARY MOCVD PROCESS



Planetary MOCVD architecture implemented in **TNL-Injector simulator** equivalent to the **AIX 200/4 horizontal MOCVD reactor**.

Inlet of the reactor is divided into two parts by a separator through which the group III and V precursors can be fed into the upper and lower inlet respectively.



PLANETARY MOCVD PROCESS



The reactant flux, $J_A = -\frac{D_{AB}}{RT} \frac{dc_A}{dx}$ with $\frac{dc_A}{dx} \approx \frac{\Delta c_A}{\Delta x} = \frac{c_{AB} - c_{AS}}{\delta}$

where J_A is the diffusion flux of specie A, c_A is the concentration of species A, x is the direction perpendicular to the substrate surface, R is the gas constant, T is the absolute temperature. δ is the chamber boundary layer thickness.

D_{AB} is the diffusivity of the bulk stream reactants and dependent on Leonard-Jones parameters (σ , Ω) based on the Chapman-Enskog theory

$$D_{AB} = 2.7 \times 10^{-3} \frac{\sqrt{T^3 \left(\frac{1}{M_A} + \frac{1}{M_B} \right)}}{p \sigma_{AB}^2 \Omega_{D,AB}}$$

M is the molecular weight, p is the pressure, σ_{AB} is the collision diameter, and $\Omega_{D,AB}$ is the collision integral and dependent on temperature and intermolecular potential.

Average boundary layer thickness, δ , $\bar{\delta} = \frac{10}{3} \sqrt{\frac{\mu_{mix} L}{\rho U}}$

PLANETARY MOCVD PROCESS



An Injector MOCVD reaction initiate either surface kinetic or mass transport control. Suppose C_g is the concentration of the bulk gas and C_s is the concentration of reactants at the substrate interface. The concentration of the reactants drops from the bulk to the substrate surface and the corresponding mass flux,

$$I_{gs} = h_g (C_g - C_s)$$

where h_g is the gas mass transfer coefficient, insensitive to variations in temperature.

The flux consumed at the surface $I_s = k_s (C_s)$ where $C_s = \frac{C_g}{1 + \frac{k_s}{h_g}}$

where k_s is the slowest surface reaction rate constant.

For $k_s \gg h_g$, the system dictated by mass controlled, low gas transport rate through the boundary layer limits the rapid surface reaction.

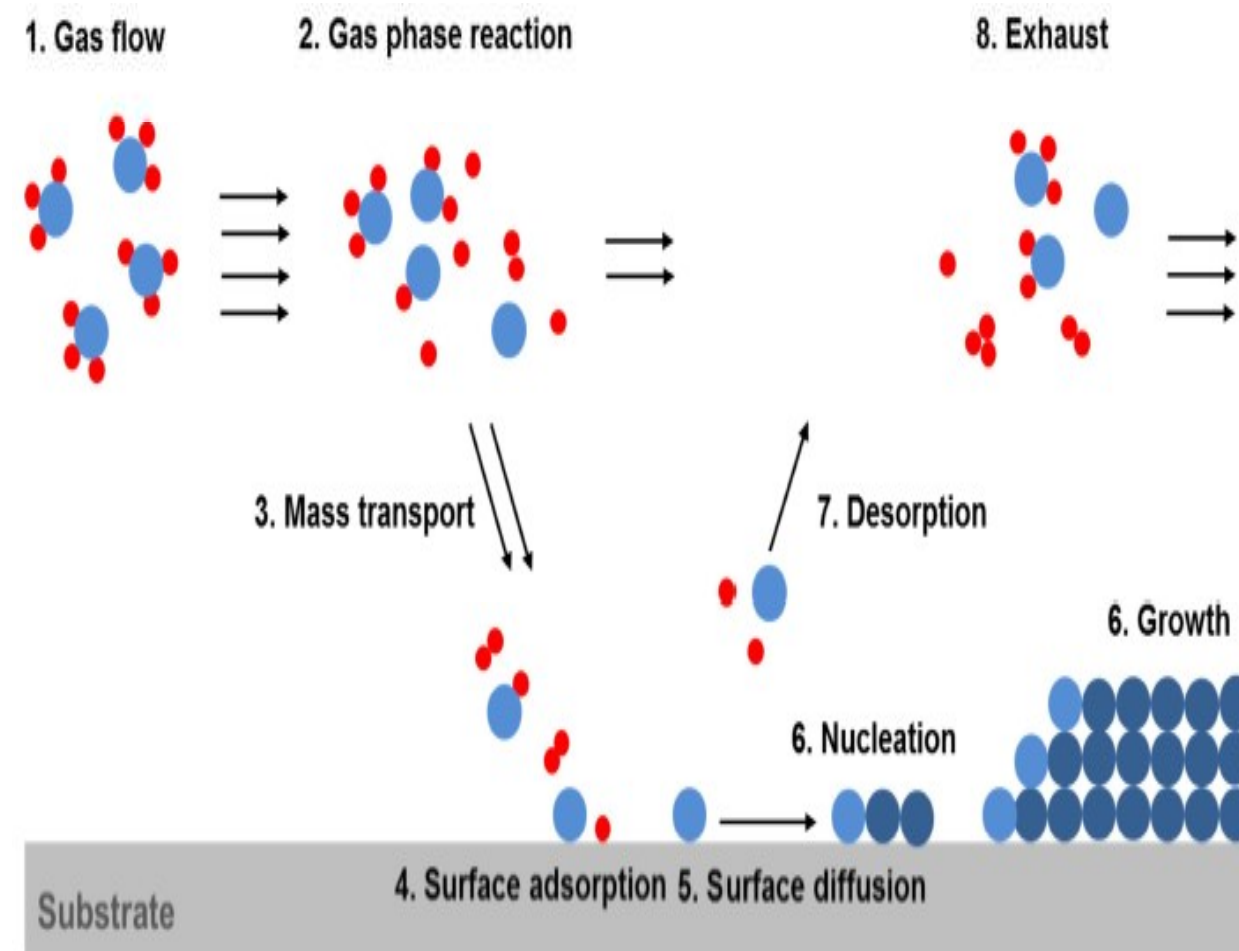
Surface reaction control dominates for $h_g \gg k_s$, the surface reaction is slow even through sufficient reactant gas is available. Additionally, h_g increases with increasing pressure and decreasing temperature and k_s follows the Arrhenius equation.



WORKING



- ❖ Gas phase kinetics
- ❖ Surface phase kinetics
- ❖ Each monolayer with atoms positions
- ❖ Defects layer by layer quantitatively and qualitatively
- ❖ Strain layer by layer
- ❖ Surface Roughness
- ❖ Lattice Constant etc.



TNL CHEMICAL DATABASE



Precursors for MN Growth

(M = Ga, Al, In)

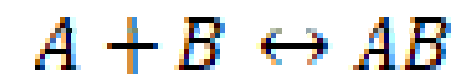
1. TMM
2. DMM
3. NH₃
4. N₂

Dopants

1. Cp₂ Mg
2. Silane

Carrier Gases: MN Growth

1. N₂
2. H₂
3. Ar



$$\frac{d[AB]}{dt} = \frac{k_1[A][B]^2}{1 + k_2[A] + k_3[B]^{1/2}}$$

Reaction rates in forward and reverse directions

$$k = AT^n \exp\left(-\frac{E_a}{RT}\right)$$

KINETIC MONTE CARLO (KMC)



Total Deposition Rate:

$$R = A + H + D$$

A - Adsorption, H – Diffusion, D - Desorption rates

$$A = Flw \quad \text{Here, } l \text{ and } w \text{ denote length and width of substrate}$$

$$h_j = D_0 \exp\left(-\frac{E_j}{k_B T}\right) \quad \text{The characteristic vibration frequency, } D_0 = \frac{2k_B T}{h}$$

$$d_j = D_0 \exp\left(-\frac{E_j^{des}}{k_B T}\right) \quad \text{with } E_j^{des} = E_S + nE_n$$

EXTRACTABLE



1. Lattice Parameters:

- Layer by layer lattice parameter Extraction.
- Averaging layer by layer lattice constant may produce overall lattice constant of film.
- The lattice constant can be calibrated with lattice constant with XRD studies.
- Lattice constant includes all the strain, defects etc effects.

2. Strain:

- Averaging layer by layer strain produce overall strain in the film.
- The strain can be calibrated with experimental strain.

3. Surface Roughness:

- Extract surface roughness as a function of growth time

$$r = \sqrt{\frac{\sum_{i=1}^N \sum_{j=1}^N [h_{ij} - \bar{h}]^2}{N \times N}}$$

Here N is the total number of lattice points, h_{ij} is the height at a given lattice point located at position i and j , on the lattice and \bar{h} is the average height of all lattice points.

4. Mole fraction:

- Extract number of atoms of different constituents layer by layer.
- Ratio of group-III & V deposited atoms → Molefraction.

5. Defects :

- Extract number of interstitials, vacancy etc layer by layer along with dislocation and Stacking Faults

CASE STUDY : Si/AlN
PLANETARY MOCVD
PROCESS

INPUT CONDITIONS

| Parameters | Si/AlN | Pre-Nitridated Si/AlN |
|-----------------------------|------------------------|------------------------|
| Chamber Temperature (°C) | 300 | 300 |
| Chamber Pressure (mbar) | 40 | 40 |
| Chamber Volume (lits) | 1.4 | 1.4 |
| Ceiling Height (cm) | 1 | 1 |
| Substrate Temperature (°C) | 1050 | 1050 |
| Precursors | TMAI & NH ₃ | TMAI & NH ₃ |
| Precursors Flow Rate (sccm) | 105 sccm & 1 slm | 105 sccm & 1 slm |
| Carrier Gas | H ₂ | H ₂ |
| PreNitridation(slm) | - | 1 |

| Parameters | Si/AlN | Pre-Nitridated Si/AlN |
|---------------------------------------|--------------------------------|-----------------------|
| Nitridation Temperature (C) | - | 1050 |
| Nitridation Time (s) | - | 30 |
| Surface Energy (eV) | 2.0 | 2.0 |
| Desorption Barrier (eV) | 3.0 | 3.0 |
| Schwoebel Barrier (eV) | 0.05 | 0.05 |
| Incorporation Barrier (eV) | 0.05 | 0.05 |
| Nearest Neighbour (eV) | 0.05 | 0.05 |
| No. of Interactive Elements | 1 | 1 |
| Substrate Dimension (A ²) | 50x50 [Unit Cell] ² | |

CHEMICAL KINETICS



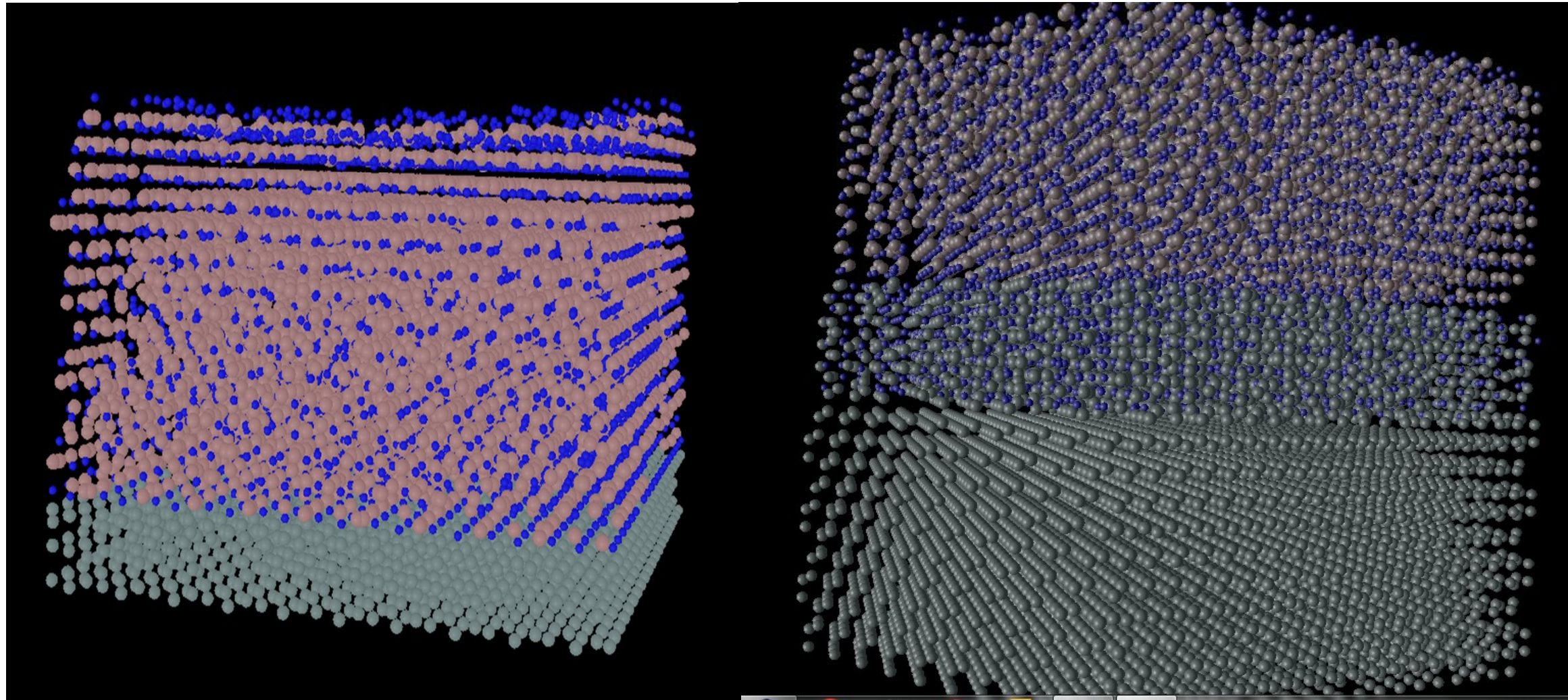
Gas Phase Equations

| | | | |
|---|-------------------------|---------|--------------|
| $\text{Al}(\text{CH}_3)_3 = \text{AlCH}_3 + 2\text{CH}_3$, | $A=3.5 \cdot 10^{15}$, | $n=0$, | $E_a= 66500$ |
| $\text{Al}(\text{CH}_3)_3 + \text{NH}_3 = \text{Al}(\text{CH}_3)_3:\text{NH}_3$, | $A=3.0 \cdot 10^{12}$ | $n=0$ | $E_a= 0.0$ |
| $\text{Al}(\text{CH}_3)_3:\text{NH}_3 = \text{Al}(\text{CH}_3)_3 + \text{NH}_3$, | $A=5.0 \cdot 10^{10}$ | $n=0$ | $E_a= 22000$ |
| $\text{Al}(\text{CH}_3)_3:\text{NH}_3 = (\text{CH}_3)_2\text{Al}:\text{NH}_2 + \text{CH}_4$, | $A=2.0 \cdot 10^{12}$ | $n=0$ | $E_a= 27000$ |
| $\text{Al}(\text{CH}_3)_3:\text{NH}_3 + \text{NH}_3 = (\text{CH}_3)_2\text{Al}:\text{NH}_2 + \text{CH}_4 + \text{NH}_3$, | $A=2.0 \cdot 10^{12}$ | $n=0$ | $E_a= 13000$ |
| $2(\text{CH}_3)_2\text{Al}:\text{NH}_2 = ((\text{CH}_3)_2\text{Al}:\text{NH}_2)_2$, | $A=4.0 \cdot 10^{11}$ | $n=0$ | $E_a= 0.0$ |

##Gas to Surface Phase Equations

| | |
|--|----------|
| $\text{Al}(\text{CH}_3)_3 + \text{space} = \text{Al}(\text{S}) + 3\text{CH}_3$, | coll 1.0 |
| $\text{Al}(\text{CH}_3)_3:\text{NH}_3 + \text{space} = \text{Al}(\text{S}) + 3\text{CH}_3 + \text{NH}_3$, | coll 1.0 |
| $\text{AlCH}_3 + \text{space} = \text{Al}(\text{S}) + \text{CH}_3$, | coll 1.0 |
| $(\text{CH}_3)_2\text{Al}:\text{NH}_2 + \text{space} = \text{AlN}(\text{S}) + 2\text{CH}_4$, | coll 1.0 |
| $((\text{CH}_3)_2\text{Al}:\text{NH}_2)_2 + \text{space} = 2\text{AlN}(\text{S}) + 4\text{CH}_4$, | coll 1.0 |

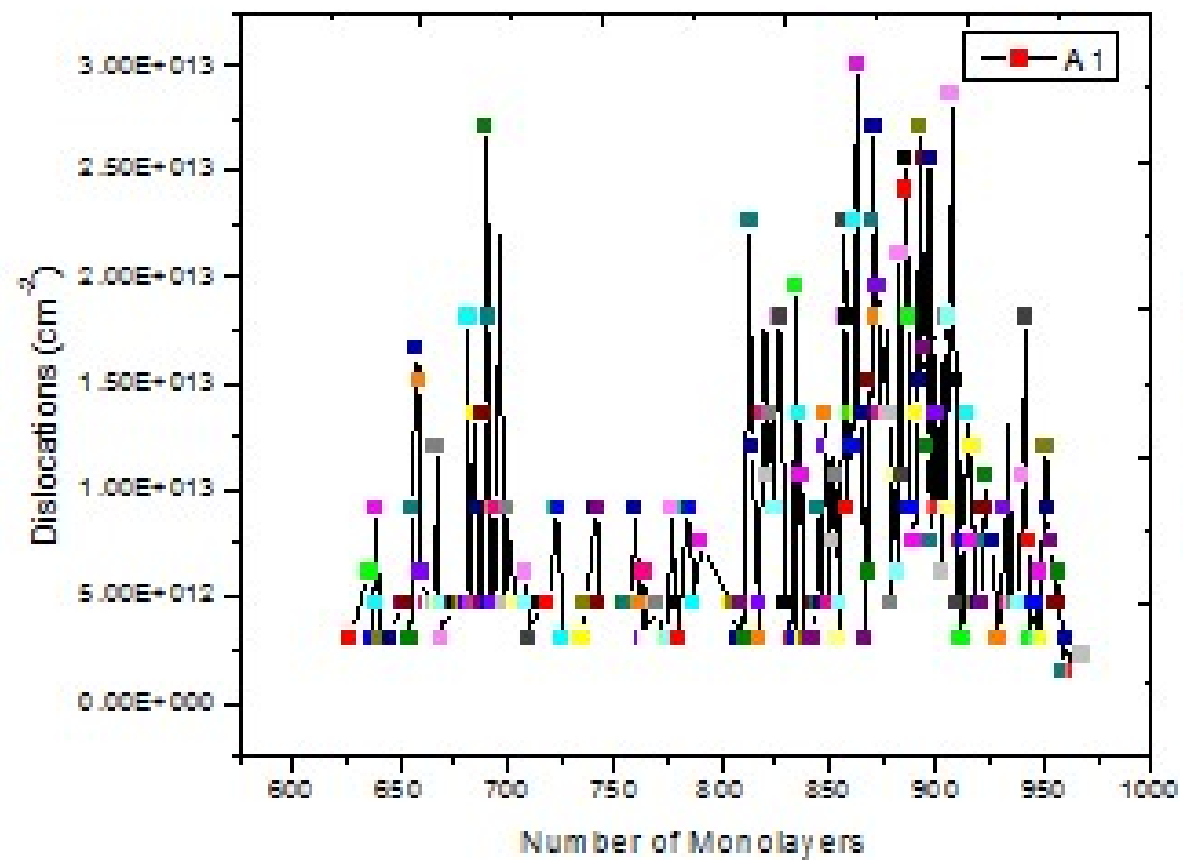
OUTPUT



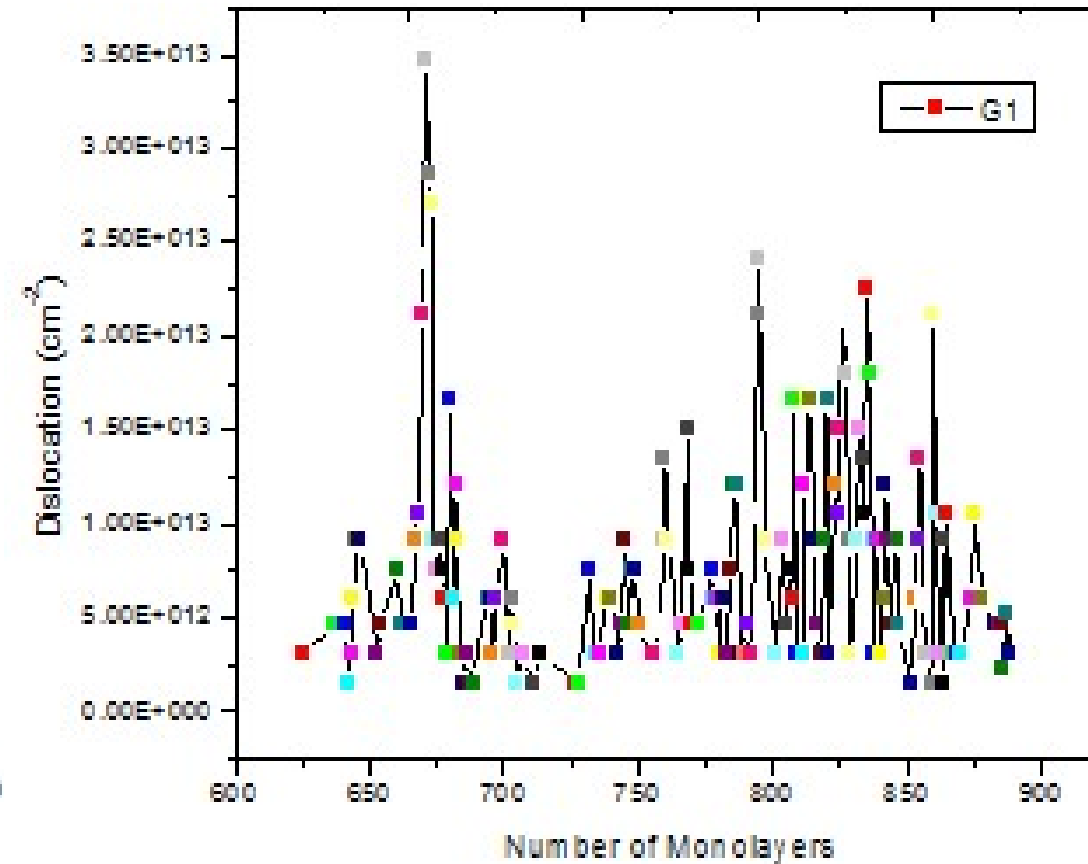
Without Nitridation

Pre-Nitridation for 30 s

DISLOCATIONS PER MONOLAYER

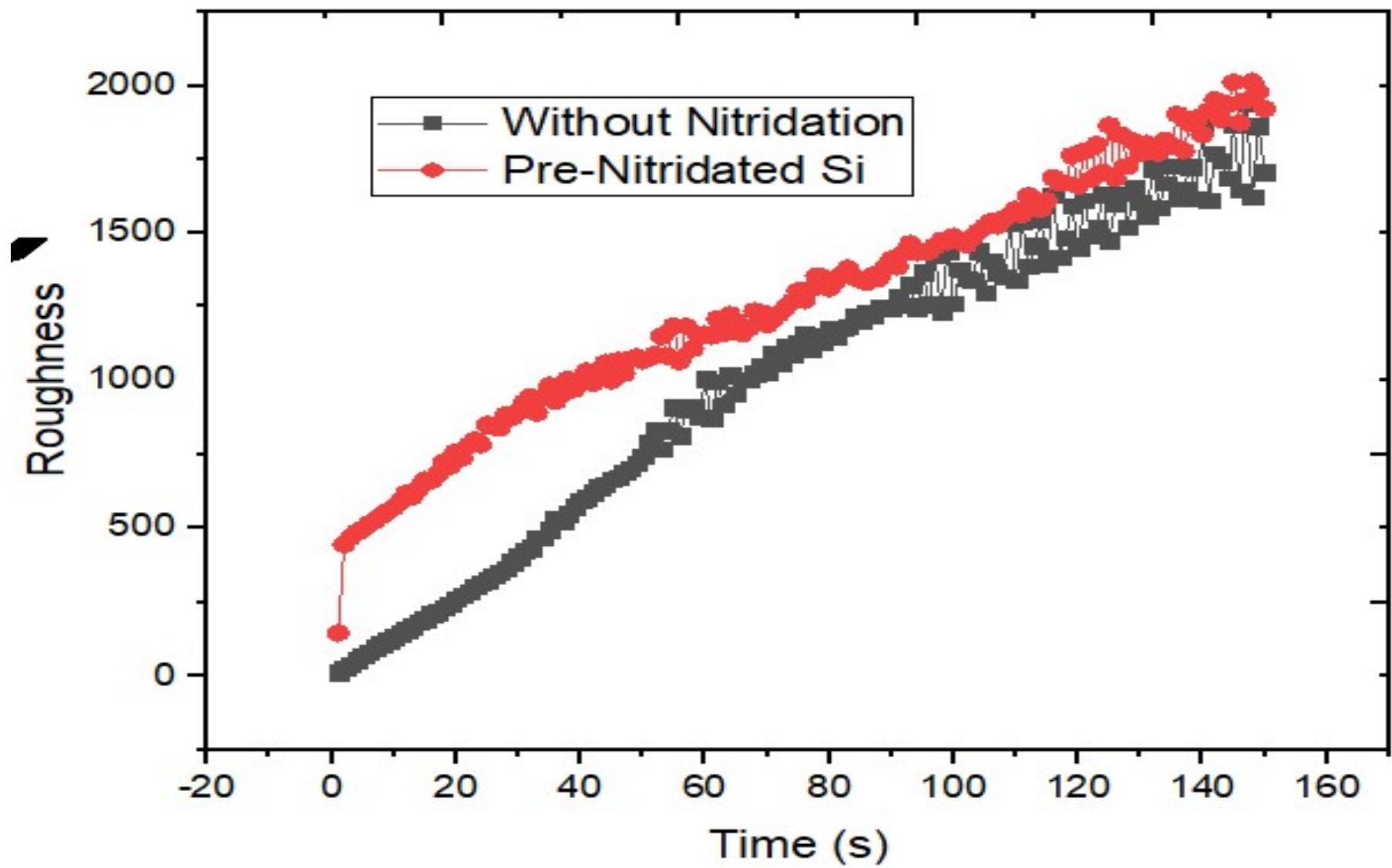


Without Nitridation



Pre-Nitridation for 30 s

SURFACE ROUGHNESS



OTHER OUTPUT

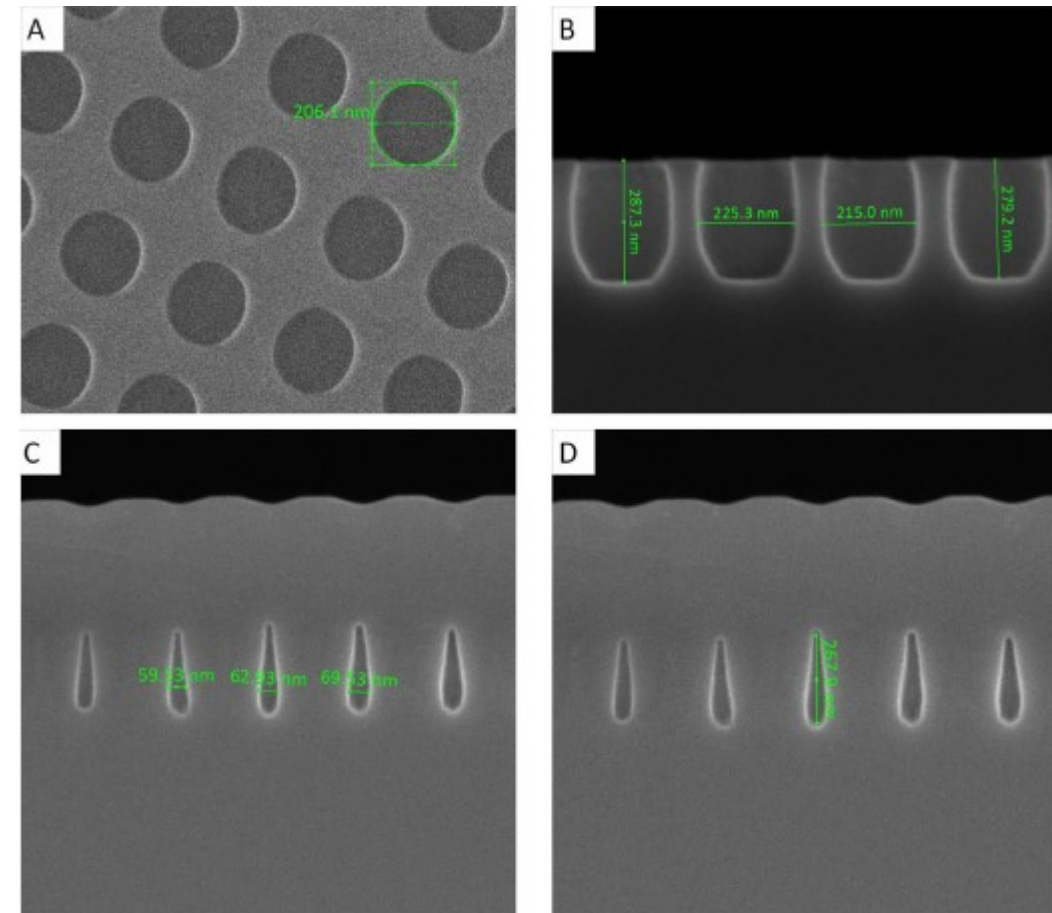
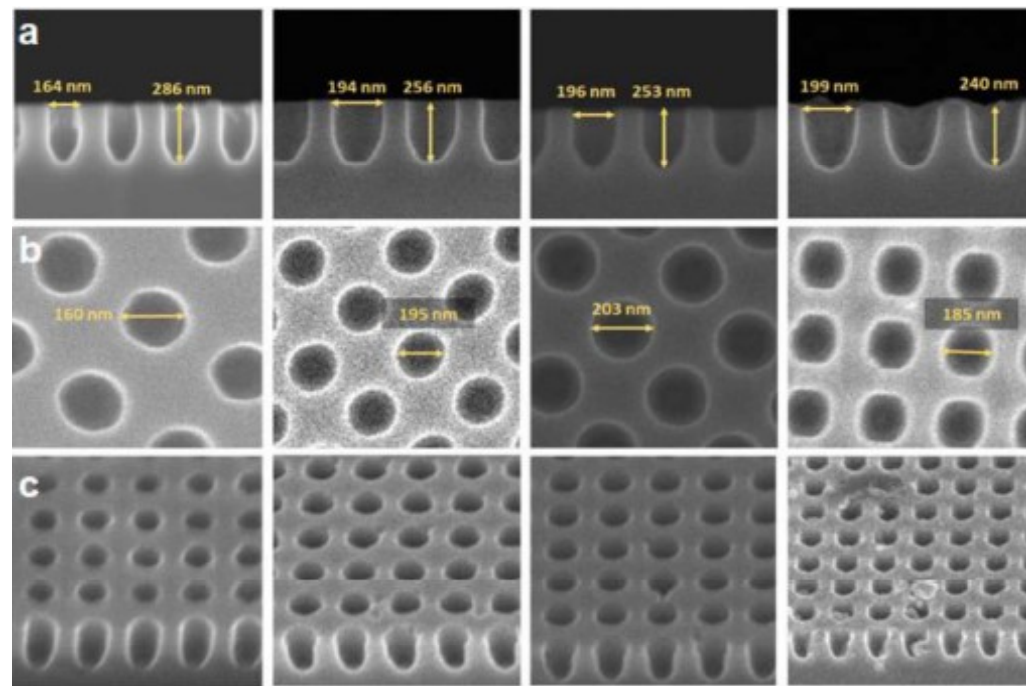
| Parameters | Si/AlN | Pre-Nitridated Si /AlN |
|--|----------------|------------------------|
| Substrate Thickness (μm) | 0.3258 | 0.3258 |
| Si ₃ N ₄ Thickness (nm) | - | 3.7817 |
| Total Deposited Atoms (AlN) | 5046178 | 5233815 |
| Vacancies (cm^{-3}) | 36069 | 13393 |
| Total Dislocation Density (cm^{-3}) | 2503 | 1612 |
| Al Atoms | $\approx 50\%$ | $\approx 50\%$ |
| N Atoms | $\approx 50\%$ | $\approx 50\%$ |
| Many More | | |

Patterned Substrate: Selective Epitaxy

TNL-Injector Simulator provides flexibilities to simulate regrowth processes at Atomistic Scale for Selective Epitaxy with capabilities:

- *Process Optimization*
- Atomistic growth process for void-semiconductor photonic crystal (PhC)
- Better understanding of *invisible Physical Phenomenon*
- *Patterned substrates Shapes: Steps, Grooves, Well etc.*
- *Epitaxial growth through: MBE, MOVPE/MOCVD*
- *Effects of regrowth on air-hole morphology*
- Comparison between patterned substrate hole regrown void's dimensions
- patterned substrate hole: *play a very critical role in the final regrowth*
- *Many More Benefits*

MAJOR CHALLENGES: DIMENSIONS OF AIR HOLE

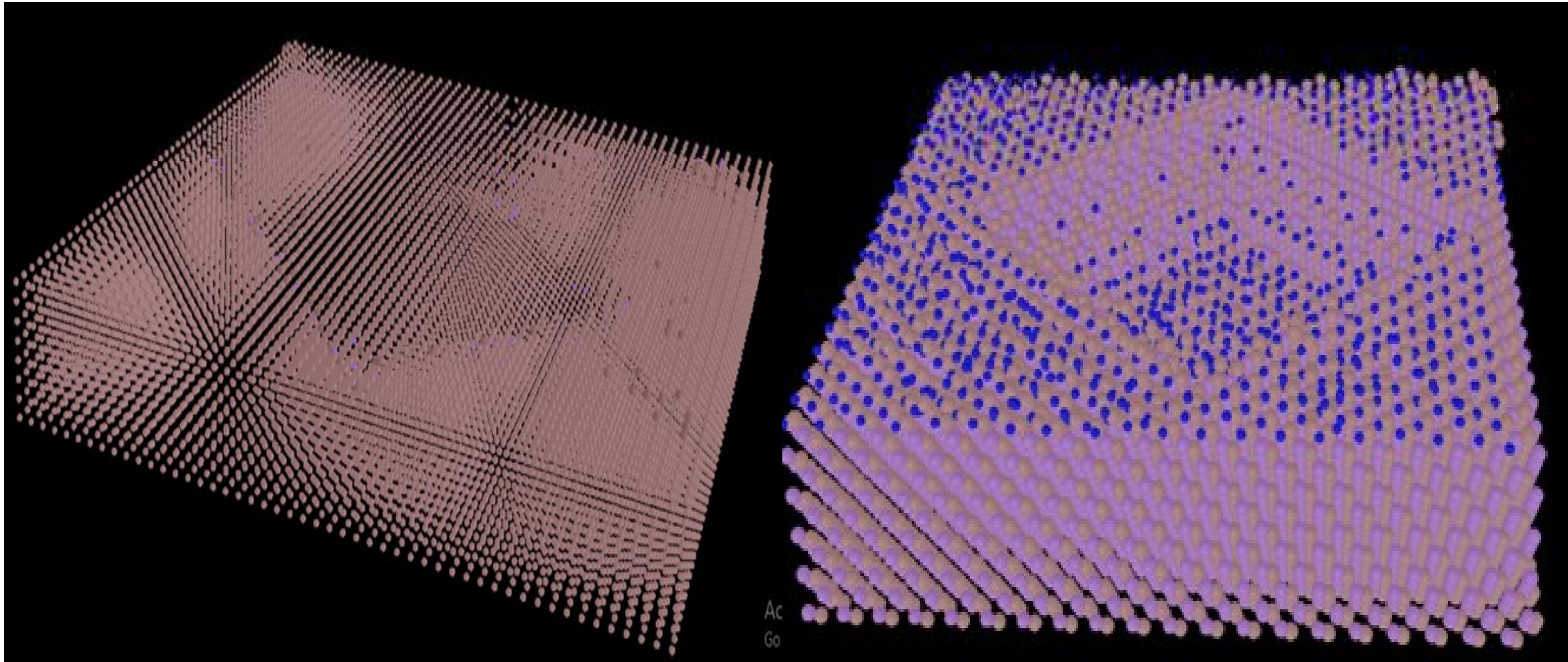


Information

Strictly Private and Confidential



Re-growth over Step Patterned Substrate



Regrowth of Si over GaAs Step Pattern Substrate
Unit Cells representation

Regrowth of Si over GaAs Step Pattern Substrate
Atomistic representation

ADVANCE LICENSING & PRICE VALUE



TNL's tools support advanced and unique licensing models tailored for unique customer needs.

➤ **ADVANCED LICENSING OPTIONS:**

- Term-Based
- Perpetual with Annual Maintenance Cost (AMC)
- TCAD Academic Suite
- 24x7 Technical Support for **Academic Institutions**



Publications



1. P.K. Saxena, numerical study of dual band (MW/LW) ir detector for Performance improvement, *Defence Science Journal*, vol. 67(2), (2017) pp. 141-148. DOI : 10.14429/dsj.67.11177
2. Praveen K. Saxena, Pankaj Srivastava, R. Trigunayat, An innovative approach for controlled epitaxial growth of GaAs in real MOCVD reactor environment, *Journal of Alloys and Compounds*, vol. 809 (2019) 151752. <https://doi.org/10.1016/j.jallcom.2019.151752>
3. Praveen Saxena, R. Trigunayat, Anchal Srivastava, Pankaj Srivastava, Md. Zain, R.K. Shukla, Nishant Kumar, Shivendra Tripathi, FULL ELECTRONIC BAND STURCTURE ANALYSIS OF Cd DOPED ZnO THIN FILMS DEPOSITED BY SOL-GEL SPIN COATING METHOD , II-VI US Workshop Proceedings, 2019.
4. R. K. Nanda, E. Mohapatra, T. P. Dash, P. Saxena, P. Srivastava, R. Trigutnayal, C. K. Maiti, Atomistic Level Process to Device Simulation of GaNFET Using TNL TCAD Tools, *Advances in Electrical Control and Signal Systems* pp 815-826, (2020), Springer Book. https://doi.org/10.1007/978-981-15-5262-5_61
5. Sanjeev Tyagi, P. K. Saxena, Rishabh Kumar, Numerical simulation of $\text{In}_x\text{Ga}_{1-x}\text{As}/\text{InP}$ PIN photodetector for optimum performance at 298 K, *Optical and Quantum Electronics* (2020) 52:374. <https://doi.org/10.1007/s11082-020-02488-1>
6. A. Srivastava, A. Saxena, P. K. Saxena, F. K.Gupta, P. Shakya, P. Srivastava, M. Dixit, S.Gambhir, R. K. Shukla & A. Srivastava, An innovative technique for electronic transport model of group-III nitrides, *Scientific Reports* 10, 18706 (2020).
7. PK Saxena, A Srivastava, A Saxena, F Gupta, P Shakya, A Srivastava, et. al., [An Innovative Model for Electronic Band Structure Analysis of Doped and Un-Doped ZnO](#), *Journal of Electronic Materials* 50 (4), 2417-2424(2021).
8. P. K. Saxena, F. K. Gupta, A. Srivastava, P. Srivastava¹ and Anshu Saxena, Ultrafast carrier's dynamics with scattering rate saturation in Ge thinfilms Ultrafast carrier's dynamics with scattering rate saturation in Ge thinfilms TechRxiv · Mar 17, 2022



Thank You
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